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# The Crystal Structure of Low Tridymite

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Low tridymite (SiO<sub>2</sub>) from Plumas County, California, is triclinic, F1, a = 9.932 (5), b = 17.216 (6), c = 81.864 (9) Å,  $\alpha = \beta = \gamma = 90^{\circ}$ ; Z = 320 [SiO<sub>2</sub>]. The structure was solved by a restrained-parameter structure-factor least-squares refinement procedure with a twinned crystal of orthorhombic diffraction aspect. The final conventional unweighted R value for 3170 reflections is 0.064. The structure can be described in terms of ten crystallographically distinct layers made up of oval rings of six linked tetrahedra. Average Si–O distances for each of the 80 tetrahedra range from 1.602 to 1.611 Å; Si–O–Si angles average 148.3° and range from 139.7 to 173.2°. There is no evidence for ordering of impurities. The structure is distinctly different from that of monoclinic low tridymite in which only one-third of the rings are oval and two-thirds are ditrigonal. Our data suggest that terrestrial low tridymite may be a lower-temperature form.

#### Introduction

Low tridymite, a polymorph of SiO<sub>2</sub>, was originally found in andesitic volcanic rocks from the Cerro San Cristóbal, Mexico (vom Rath, 1868). Recently Gardner & Appleman (1974) have shown that these crystals, as well as tridymites from many other natural terrestrial occurrences, have the orthorhombic diffraction aspect  $F^{***}$  with approximate cell dimensions a = 9.9, b = $17 \cdot 1, c = 81 \cdot 6$  Å and compositions close to  $98 \cdot 5$  at.% SiO<sub>2</sub>. This cell differs significantly from that of monoclinic 'low tridymites' from-meteorites and synthetic preparations (Dollase, 1967) and lunar rocks (Appleman, Nissen, Stewart, Clark, Dowty & Huebner, 1971; Dollase, Cliff & Wetherill, 1971) which have approximate dimensions a = 18.5, b = 5.0, c = 23.8Å,  $\beta = 105 \cdot 7^{\circ}$ , space group Cc (Dollase, 1967). The structure of the monoclinic form has recently been reported (Dollase & Baur, 1976; Kato & Nukui, 1976).

The structure of a terrestrial low tridymite from Plumas County, California, is described in this paper.

This investigation was originally begun because the suggested similarities between the framework topologies of tridymite and silica glass (Konnert & Karle, 1972) made a more precise knowledge of the tridymite structure desirable. A second goal of the work was to determine whether the small amounts of Al, K and Na usually present in tridymites played an essential role in stabilizing the structure. Thirdly, we wished to elucidate the structural differences between the terrestrial and meteoritic forms. A preliminary account of this study was presented to the 1975 Annual Meeting of the Mineralogical Society of America (Konnert & Appleman, 1975).

## Experimental

The crystal selected for structure analysis in this study was obtained from a rhyolitic volcanic rock from Plumas County, California, in the Collections of the Smithsonian Institution, National Museum of Natural History specimen No. 121899. Tridymite from this locality shows less twinning than most natural specimens. The particular crystal which we used showed no evidence of pseudo-hexagonal twinning on precession and Weissenberg X-ray diffraction photographs. Zero- and upper-level nets were compatible with diffraction aspect  $F^{***}$ , orthorhombic (Gardner & Appleman, 1974). All reflections registered as sharp spots, with only a little very weak diffuse streaking parallel to  $c^*$ .

Several crystals from the Plumas County specimen were analyzed with an ARL SEMQ electron microprobe, with standard techniques. Many points on each crystal were analyzed individually. The results show an average composition  $(Si_{0.99}Al_{0.01}K_{0.002}-Na_{0.003}Ti_{0.001})O_2$ . There is no significant variation within crystals, or from crystal to crystal. Thus, the analyzed material is approximately 99 mol% SiO<sub>2</sub>.

X-ray diffraction data were collected on a Picker FACS-I diffractometer with a 1° min<sup>-1</sup> scan speed and 20 s counting time on each background, with Ni-filtered Cu Ka radiation. Special collimators were fitted to reduce background. No deviations from orthorhombic symmetry and no evidence of split reflections were observed. Lattice parameters, determined by least-squares refinement of  $2\theta$  angles for selected single-crystal reflections measured on the FACS-I, are: a = 9.932 (5), b = 17.216 (6), c = 81.864 (9) Å for the orthorhombic F cell. 3170 diffraction maxima were collected to  $2\theta = 126^{\circ}$ .

The X-ray diffraction intensity data were corrected for Lorentz and polarization effects. Absorption corrections were made with the programs of Stewart, Kruger, Ammon, Dickinson & Hall (1972),  $\mu$ (Cu  $K_{\alpha}$ ) = 78.64 cm<sup>-1</sup>. The volume of the irregular crystal fragment used in the study was 0.002 mm<sup>3</sup>. A single isotropic extinction parameter (Zachariasen, 1967) was refined throughout the investigation.

#### Determination of the trial structure

Precession and Weissenberg photographs indicated the diffraction symmetry to be *Fmmm*. Such symmetry for a single crystal would limit the possible space groups to F222, *Fmm2*, or *Fmmm*. However, inspection of the *hk*O Patterson map revealed that the projected electron densities for the 20 layers were approximately superimposed, were incompatible with the above space groups, and compatible with plane group *pg*. In fact, with exact overlap of the layers assumed, refinement (with isotropic *B*'s) of the two Si atoms and the three O atoms of plane group *pg* resulted in a conventional unweighted R = 0.19. The impossibly high *B* values of 3–4 for the Si atoms and 5–6 for the O atoms indicated, not surprisingly, that the assumption of layer overlap is not strictly valid. The *R* of 0.19 did, however, strongly suggest that it would not be fruitful to proceed under the assumption that the space group was F222, Fmm2 or Fmmm.

All values of |E| > 1.08 (~1500) were associated with the reflections of parity ggg. Further, all values of |E| > 2.14 (~140) were associated with a diffraction pattern of symmetry mmmP-n- with a halving of the cell dimensions. Space group P2<sub>1</sub>nm, with hk0 projection pg, is consistent with this diffraction pattern. A phase determination was carried out in space group P2<sub>1</sub>nm with the symbolic addition procedure (Karle & Karle, 1966), and a resulting E map indicated a possible framework of Si atoms. The initial trial structure was obtained by placing the O atoms midway between the bridged Si atoms.

## Refinement of the structure

Refinement of the trial structure was carried out in several stages. As mentioned, all values of |E| > 1.07 were of parity ggg. The first stage of the refinement was carried out with the ggg data in space group  $P2_1$  with the unique direction coinciding with the *a* axis of the orthorhombic cell. The *hk*0 projection of  $P2_1$  has symmetry pg. A twin plane perpendicular to **c** was assumed such that  $I_c(hkl) = I'_c(hkl) + I'_c(hk\bar{l})$ . Several cycles of refinement reduced the R value from 0.40 to 0.22 for the ggg data. The R value for the *uuu* data was, of course, 1.00 with an overall R of ~0.40.

A probable cause for the orthorhombic diffraction was the presence of more than one type of twinning. In order to investigate such possibilities, each intensity was expressed as the sum of four calculated intensities:

$$I_{c}(hkl) = k_{1}I_{c}'(hkl) + k_{2}I_{c}'(hkl) + k_{3}I_{c}'(hkl) + k_{4}I_{c}'(hk\bar{l}).$$

Intensities of the *i*th twin component were scaled with  $k_i$ . The refinement was then carried out in space group F1 with 240 atoms in the asymmetric unit. A point to note is that all four  $I_c$  components of each observation may refine to the same value if the true symmetry is orthorhombic. They may refine to pairs of equal values if the true symmetry is monoclinic, or they may, as turned out to be the case, all be different if the symmetry is triclinic. If the assumption of complete overlap (each observation being the sum of four intensities) were not correct, it should not be possible to attain a suitably low R value.

A restrained-parameter structure-factor least-squares refinement procedure was formulated at this stage. The technique is described elsewhere (Konnert, 1976) and will only be discussed briefly here. It combines the conditional structure-factor least-squares procedure described by Waser (1963) with the conjugate gradient (CG) method for solving linear systems (Hestenes & Stiefel, 1952). The sum of squared residuals to be All values are fractional coordinates of the F cell multiplied by 10<sup>5</sup>. Si atoms are identified by a layer number (in parentheses) and an atom number (1-8); the atoms are numbered in the same way in each layer. O atoms within a layer are described in the same way as the Si atoms; O atoms shared between layers have both layer numbers in parentheses.

ATOM	X	Y	2	ATOM	X	Y	Z	ATOM	x	Y	Z
SI( 1)1	6049	3633	3414	0( 1,2)1	2331	2880	5321	1 0( 4) 5	-6677	13958	17288
SIC 1)2	56126	3679	3371	0( 1,2)2	5 32 02	3557	5282	0(4)6	53602	10036	17960
SIC 133	-2022	19793	1846	0(1,2)3	18855	30450	4704		-13692	28276	17792
SIC 177	48204	19712	1779	0(1,2)	-6621	22910	9697	0(4)0	36136	29022	17220
SIC 1)6	72606	29128	2822	0( 2,3)2	45120	22872	9819	0( 4)10	60790	24769	18233
SIC 1)7	31136	46063	1895	0( 2,3)3	28736	43393	10126	0( 4)11	29727	35425	17561
SIC 1)8	81084	46158	1933	0( 2,3)4	78692	42769	10194	0( 4)12	67902	38824	17331
51( 2)2	1300	3039	7135	0( 3,4)1	53586	6824	14807		-48/9	-1691	21988
SIC 213	-1645	22688	7960	0( 3.4)3	19016	26991	15158	0(5)2	45224	-2256	22300
SIC 234	23070	31107	6598	0( 3,4)4	69419	27038	15271	0( 5) 4	70138	2185	22520
SIC 2)5	46080	20317	7936	0( 4,5)1	-6233	19435	20250	0( 5) 5	1610	12601	22966
51( 2)6	71072	28419	6591	0(4,5)2	43292	18790	20259	0(5)6	50938	12330	2 30 2 4
SIC 2)8	73949	44146	8340	0( 4.5)4	79075	46998	19732	0(5) 7	7875	27201	22448
SIC 3)1	5429	4645	12921	0( 5,6)1	3607	890	24997	0( 5) 9	32898	23136	23066
SIC 3)2	55652	4513	12951	0( 5,6)2	54753	924	25073	0( 5)10	57759	26756	22387
SIC 333	-2557	21178	11701	0( 5,6)3	20233	32513	25050	0( 5)11	26947	37294	22080
SIC 3)5	48158	21099	11717	0( 5,8)4	-6081	19694	29712		-13495	-1919	22090
SI( 3)6	72795	28732	1 3 3 7 5	0( 6,7)2	43250	19703	29745	0( 6) 2	11728	1166	28050
SIC 3)7	30672	45008	12047	0( 6,7)3	28913	46581	30342	0( 6) 3	36767	-2723	27257
SIC 338	80469	45095	12077	0( 6,7)4	78507	46136	30288	0(6)4	61405	1162	28149
51( 4)1	51258	3122	16666	0(7,8)1	54 099	6641	35209		-4016	12214	26904
SIC 433	- 34 31	21530	18369	0( 7.8)3	18331	26704	34864	0(6)7	-23071	22311	27351
SIC 4)4	24039	27072	17021	0( 7,8)4	69788	26518	34786	0( 6) 8	1841	26978	27030
SIC 4)5	48592	18696	18403	0( 8,9)1	-6560	22485	40227	0( 6) 9	26732	22413	27369
51( 4)6	71198	29733	17140	0( 8,9)2	44709	22556	40200	0( 6)10	51778	26532	27027
51( 4)8	73858	47061	17894	0( 8,9)4	79172	43183	39803	0( 6)12	69288	36573	28107
SIC 5)1	4792	3540	23125	0(10,9)1	2675	2229	44667	0(7)1	-13170	-455	33348
SIC 5)2	54997	3394	23180	0(10,9)2	53965	2918	44729	0(7)2	11483	3043	32267
51( 5)3	-3233	20609	22157	0(10,9)3	19695	30664	45287		36017	74	33242
SIC 5)5	46402	20260	22158	0(10,1)1	-5068	17436	49912		-6702	13839	32192
SIC 536	71696	29734	23155	0(10,1)2	45126	17230	49990	0(7)6	53016	10446	31991
SIC 5)7	29700	46113	21579	0(10,1)3	28663	48885	50083	0(7)7	-13852	28144	32215
510 538	79829	45915	21663	0(10,1)4	78581	48869	50049	0(7)8	11564	24608	31821
SIC 6)2	49360	2861	26878		21504	1065	3152	0(7)9	50017	21043	31821
51( 6)3	-7624	20251	27784	0(1)3	46516	-2383	2440	0( 7)11	29305	35586	32543
SIC 6)4	17311	29695	26892	0(1)4	71548	966	3108	0( 7)12	68387	38747	32871
SIC 635	42106	20037	27810	0(1)5	3628	12458	2849	0(8)1	-4270	-2487	37553
51( 6)6	24310	29420	28464		-15552	22774	2711		20763	1843	37297
51( 6)8	74099	45386	28401	0(1)8	9752	26258	1843	0(8)4	70792	1982	37353
SIC 7)1	-1127	5741	33366	0(1)9	34720	22544	2695	0(8)5	2510	12080	38207
S1C 7)2	51099	3461	33274	0( 1)10	59632	26374	1782	0(8)6	52151	12008	38091
SIC 734	23592	21525	33012		28313	36924	2027		9630	22357	37864
SIC 7)5	48141	19216	31597	0(2)1	-13521	3536	7910	0(8)9	34475	22541	37221
SIC 736	71254	29525	32940	0(2)2	12074	451	8279	0( 8)10	59553	26715	37730
51(7)7	26507	44681	32239	0(2)3	36614	-3305	7245	0( 8)11	27131	37337	36860
STC 831	5402	46003	37071	0(2)4	4317	14369	7451		-13518	2894	42042
SIC 852	55433	4489	37002	0(2)6	42613	11183	7778	0 9 2	12484	585	41725
SIC 8)3	-2367	20997	38348	0(2)7	-13907	25131	6786	0(9)3	37026	-3506	42775
51(8)4	22606	28414	36728	0(2)8	10255	29035	7728	0( 9) 4	61250	-547	41758
SIC 8)6	72917	20929	36679		60949	23204	7209	0(9)5	43439	14034	42652
SIC 8)7	30406	44985	37919	0( 2)11	27766	39862	7004	0 9 7	-13450	25449	43264
8(8)12	80457	44962	37903	0( 2)12	68080	36035	7652	0(9)8	10837	28407	42256
SIC 9)1	1089	4822	42801	0(3)1	- 3956	-2582	12417	0(9)9	35841	24435	43203
SIC 912	-1648	22402	*2809 42095		20800 45876	-2355 -2388	12346	0( 9)10	0145U 28704	21651	42794
SIC 9)4	23522	30330	43401	0 33 4	71016	1902	12657	0( 9)12	68536	35960	42246
SIC 9)5	46602	20096	42080	0(3)5	1672	12187	11855	0(10) 1	-11012	-2326	47172
SIC 9)6	71384	28565	43382	0(3)6	53102	12274	11909	0(10) 2	13973	2249	47529
51( 9)8	20117	42818	41648		-12123	25020	12863	0(10) 3	38978	-2228	47128
SI(10)1	89	3822	46580		34668	22426	12767	0(10) 5	-4298	2300	46855
SI(10)2	50365	3917	46628	0( 3)10	59670	27066	12288	0(10) 6	45817	12756	46933
SI(10)3	-6794	19996	48037	0( 3)11	27857	37 158	13039	0(10) 7	-22182	22704	47767
51(10)4	18252	29559	47201		17435	37635	13196	0(10) 8	2843	27144	47573
SI(10)6	68040	29548	47172	00452	11944	3418	17769	0(10)10	52864	27213	47694
SI(10)7	25501	46482	48221	0( 4) 3	36308	-436	16748	0(10)11	20922	37595	48142
SI(10)8	75445	46519	48184	0(4)4	61930	-3701	17025	0(10)12	71419	37524	48105

minimized,  $\theta$ , is a function not only of the observed and calculated intensities, but also of the ideal and calculated distances. The form of  $\theta$  is:

$$\theta = \sum_{i} w_{i} (|F_{o}|_{i} - |F_{c}|_{i})^{2} + \sum_{i} w(d_{i}^{\prime 2} - d_{c,i}^{2})^{2}$$

where *i* may range over all or just a portion of the intensity data, and *l* ranges over the distances to be restrained. An ideal distance is designated as d' and a calculated one as  $d_c$ . The weight assigned to an observation is *w*. Rapid, meaningful convergence may be obtained by calculating only those elements in the matrix of the normal equations that are related to the restraint equations. This corresponds to about 4% of one half of the symmetric matrix for the case of interest. The derivative matrix is not inverted and remains unaltered during solution of the normal equations by the CG method. Thus, the advantages of a sparse matrix are retained.

The number of structural parameters to be determined by the intensity data is greatly reduced by restraining the bond distances.

Correlations of the shifts, due to the make-up of the matrix, reduce the problem to one largely of torsions. Therefore, a much smaller intensity subset is required than would be the case without the restraints. Since calculation of the structure factors and their derivatives is the time-consuming step of a refinement cycle, employing a limited number of structure factors facilitates the evaluation of different models. The 320 Si–O distances were restrained to be near 1.61 Å, the 480 O–O distances of the tetrahedra to 2.63 Å and the 160 shortest Si–Si distances to 3.08 Å.

When atomic positions are only approximately known, it is advantageous to use a subset of intensity data consisting of a low-angle shell of data. The lowerangle data correspond to large interplanar spacings, and, therefore, the derivatives involved in the Taylor expansion on the structure factors are valid over a greater range of atomic coordinates. Of the 3170 diffraction data collected, the initial cycles were carried out with the ~600 data with  $(\sin \theta/\lambda) < 0.33$ . The centroid of the molecule was constrained with three Lagrangian multipliers, *i.e.* the sum of the shifts of the x coordinates was constrained to zero as were the sums of the y and z shifts. Four thermal parameters were used to describe the system. All Si atoms were given the same isotropic B. A single anisotropic ellipsoid was used to describe the O atoms. The orientation of the ellipsoid for each O atom was fixed by the atomic environment of the atom. One axis was fixed parallel to the line connecting the bridged Si atoms, and one axis perpendicular to the Si-O-Si plane. With the inclusion of four scale factors, three Lagrangian multipliers, and one isotropic extinction parameter, 732 parameters were simultaneously varied on each cycle. After several cycles with  $w_i = 1/75^2$  and  $w = (6/d')^2$ , the conventional R was 11% for the 600 data refined and 13% for all data. Subsequent cycles, with  $w_i = (1/75)^2$ ,  $w_l(Si-O) = (8\cdot4/d')^2$ ,  $w_l(O-O) = (3\cdot1/d)^2$  and  $w_l(Si-Si) = 1\cdot0$  reduced R to 8.9%.

A difference Fourier map was computed at this stage. Because of the twinning, this was not a conventional difference map. A decision had to be made as to how  $\Delta I$  would be apportioned among the four intensities involved. From among the infinite number of ways of doing this, one was chosen where the differences for the four reflections were set equal:

$$I_o = \sum_{i=1,4} (|F_c|_i + \Delta)^2$$
$$\Delta = \{ [(\Sigma |F_c|)^2 + 4(I_o - \Sigma |F_c|^2)]^{1/2} - \Sigma |F_c| \} / 4.$$

The plus sign for the square root was chosen so as not to change the sign of  $|F_a|$ . The  $\Delta$  so determined was then assigned each of the four sets of indices and phases to obtain the Fourier map. The largest peaks were around framework atoms in what will be described later as layer No. 9. The map indicated that four O atoms and four Si atoms were misplaced by rotations about certain O-O directions. That the structure of the framework is such as to permit a gross rearrangement of some of the atoms without disrupting the stereochemistry imposed is interesting in its own right and will be discussed later. Repositioning of these atoms and further refinement reduced R to 6.4%. The largest peaks in a difference Fourier map computed at this stage were around Si atoms, suggesting that the electron density around these atoms is somewhat anisotropic. Refinement was terminated at this stage. The final coordinates are listed in Table 1.\*

Because of the extraordinarily large number of structural parameters relative to the number of independent intensity data, we feel that further unconstrained least-squares refinement would not have yielded meaningful results. The next logical step, should further refinement of this data be undertaken, would be to allow for anisotropic thermal factors for Si atoms. A still further step might be to separate the O atoms into two populations for thermal refinement: in-layer and between-layer. In monoclinic low tridymite these two groups have significantly different thermal parameters (Baur, 1977).

#### **Discussion of structure**

The 320 Si–O distances range from 1.585 to 1.623 Å with an average of 1.606 Å. They are listed in Table 2.

<sup>\*</sup> A list of structure factors has been deposited with the British Library Lending Division as Supplementary Publication No. SUP 33016 (65 pp.). Copies may be obtained through The Executive Secretary, International Union of Crystallography, 13 White Friars, Chester CH1 1NZ, England.

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# Table 2. Si-O tetrahedral distances in low tridymite (Å)

Si atoms are those from Table 1; O atoms are related to those of Table 1 by the symmetry transformations of Table 5. The average Si-O distance is given for each tetrahedron. Estimated standard deviation of an individual distance is approximately 0.005 Å.

ATON 1	ATON 2	SYM2	DIS	ATOM	2 SYM2	DIS	AVG	ATOR 1	ATON 2	STR2	DIS	ATON 2	SYM2	DIS	AVG
SFC 1)1	0( 1,2)1	1	1.609	0( 1)	1 1	1.614	1.610	\$1( 6)1	0( 5,6)1	1	1.606	0( 6) 1	1	1.603	1.604
SIC 1)2	0(1)2	1	1.591	0(1)	3 1	1.603	1.602	SI( 6)2	0( 5,6)2	1	1.608	0(6)3	i	1.608	1.607
\$1(-1)3	0(1)4	1	1.616	0(1)	6 1	1.596	1.610	516 633	0(6)4	1	1.612	0(6)6	1	1.599	1.604
	0(1)7	i	1.603	0(1)	<b>8</b> 1	1.614			0( 6) 7	i	1.614	0( 6) 8	i	1.614	
51( 1)4	a(1,2)3 a(1) 9	1	1.609	0(1)	8 1 11 1	1.616	1.610	51( 6)4	0( 5,6)3	1	1.610	0(6)8	1	1.610	1.611
SIC 1)5	0(10,1)1	9	1.611	0(1)	6 1	1.601	1.609	SI( 6)5	0( 6,7)2	1	1.589	0( 6) 6	1	1.601	1.605
SIC 1)6	3( 1,2)4	i	1.607	0(1)	7 2	1.609	1.607	SI( 6)6	0( 5,6)4	i	1.609	0( 6) 7	ż	1.609	1.608
SIC 1)7	0(10,1)4	11	1.607	0(1)	1 6	1.597	1.609	SIC 6)7	0( 6,7)3	1	1.612	0( 6) 1	6	1.598	1.608
51( 1)8	0(1)4	7	1.614	0(1)	11 1	1.602	1.606	51( 6)8	0( 6) 4	7	1.610	0(6)11	1	1.611	1.607
	0(1)3	6	1.608	0(1)	12 1	1.595	1 ( 0 0		0( 6) 3	6	1.602	0( 6)12	i	1.609	
31(2)1	0(2)2	1	1.609	0(2)	5 1	1.613	1.000		0(7)2	i	1.611	0(7)5	1	1.612	1.070
\$1( 2)2	0( 1,2)2	1	1.587	0(2)	3 1	1.616	1.605	SI( 7)2	0( 7,8)2	1	1.609	0(7)3	1	1.608	1.610
SIC 2)3	0( 2,3)1	1	1.608	0( 2)	5 1	1.605	1.610	SI( 7)3	0( 6,7)1	1	1.599	0( 7) 5	1	1.605	1.604
51( 2)4	0(1,2)3	1	1.610	0(2)	8 1	1.613	1.610	51(7)4	0(7,8)3	1	1.606	0(7)8	i	1.604	1.607
\$1( 2)5	0(2)9	1	1.603	0(2) 0(2)	11 1 6 1	1.612	1.609	51( 7)5	0(7)9	1	1.616	0(7)11	1	1.604	1.605
616 236	0(2)9	1	1.605	0(2)	10 1	1.609	1 409	577 724	e( 7) 9	1	1.607	0( 7)10	1	1.600	1 4 0 0
31( 270	0( 2)10	1	1.621	0(2)	12 1	1.601	1.000	31( 1)0	0( 7)10	i	1.601	8( 7)12	1	1.614	1.008
51( 2)7	0(2,3)3	17	1.615	0(2)	16 111	1.607	1.611	SI( 7)7	0( 6,7)3	17	1.604	0( 7) 1 0( 7)11	6 1	1.605	1-607
SIC 2)8	0( 2,3)4	1	1.607	0(2)	26	1.604	1.608	SIC 7)8	0( 6,7)4	1	1.603	0(7)2	6 4	1.615	1.609
51( 3)1	7( 3,4)1	1	1.596	0(3)	1 1	1.609	1.606	51( 8)1	0( 7,8)1	ĩ	1.600	0(8)1	i	1.606	1.606
51( 3)2	0(3)2 0(3,4)2	1	1.611	0(3) 0(3)	5 1 3 1	1.608	1.606	51( 8)2	0(8)2 0(7,8)2	1	1.610	0(8)5 0(8)3	1	1.611	1.607
\$17 213	0(3)4	1	1.609	0(3)	6 1	1.606	1.608	517 833	0(8)4	1	1.611	0(8)6	1	1.605	1.611
	0(3)7	i	1.603	0( 3)	8 1	1.606	1.000		0(8)7	i	1.608	0(8)8	i	1.606	
51( 3)4	0(3,4)3	1	1.610	0(3)	8 1	1.606	1.607	51( 8)4	0( 7,8)3	1	1.605	0(8)8	1	1.613	1.698
\$1( 3)5	0(2,3)2	1	1.611	0(3)	6 1	1.604	1.608	SI( 8)5	0( 8,9)2	1	1.617	0(8)6	1	1.605	1.608
51( 3)6	0( 3,4)4	i	1.615	0( 3)	7 2	1.605	1.608	51( 8)6	0( 7,8)4	i	1.611	0( 8) 7	2	1.610	1.607
SIC 3)7	0( 2,3)3	1	1.604	0(3)	12 1	1.607	1.607	51( 8)7	0( 8,9)3	1	1.605	0(8)12	1 6	1.600	1.607
51( 3)8	0(3)4	7	1.605	0(3) 0(3)	11 1	1.601	1.607	51( 8)8	0(8)4 0(8.9)4	7	1.605	0(8)11 0(8)2	1	1.610	1.605
\$16 431	0(3)3	6	1.609	0(3)	12 1	1.606	1 404		0(8)3	6	1.614	0( 8)12	1	1.609	1 407
51( 471	0(4)2	1	1.607	0( 4)	5 1	1.612	1.000	31( 7)1	0( 9) 2	1	1.609	0(9)5	i	1.612	1.001
51( 4)2	( 3,4)2 ( 4) 4	1	1.613	0(4)	3 1 6 1	1.608	1.610	\$1( 9)2	0(10,9)2 n( 9) 4	1	1.595	0(9)3	1	1.623	1.607
SI( 4)3	0( 4,5)1	1	1.606	0(4)	5 1 B 1	1.608	1.608	51( 9)3	0( 8,9)1	1	1.605	0(9)5	1	1.603	1.607
SI( 4)4	0( 3,4)3	i	1.605	0( 4)	8 1	1.609	1.610	SI( 9)4	0(10,9)3	i	1.591	0( 9) 8	i	1.605	1.693
51( 4)5	0(4)9	1	1.618	0(4)	6 1	1.607	1.610	51( 9)5	n( 8,9)2	1	1.598	0( 9) 11	1	1.620	1.607
51( 4)6	0(4)9	1	1.614	0(4)	10 1	1.606	1.610	51( 9)6	0(9)9	1	1.595	0( 9)10	1	1.609	1.607
	0( 4)10	i	1.612	0( 4)	12 1	1.607			0( 9)10	i	1.619	0( 9)12	ĩ	1.602	
51( 4)/	0(4)4	7	1.611	0(4)	11 1	1.607	1.606	510 977	0( 8,9)3	7	1.608	0(9)1	1	1.602	1.609
SI( 4)B	0(4,5)4	1	1.591	0(4)	26	1.615	1.605	51( 9)8	0( 8,9)4	1	1.596	0(9)2	6	1.605	1.606
STC 531	0( 5,6)1	1	1.603	0( 5)	1 1	1.613	1.606	51(10)1	0(10,9)1	1	1.611	0(10) 1	ī	1.603	1.605
51( 5)2	0( 5,6)2	1	1.607	0(5)	3 1	1.607	1.605	\$1(10)2	0(10,9)2	i	1.604	0(10) 3	1	1.602	1.604
51( 5)3	0( 5) 4 0( 4,5)1	1	1.611	0(5) 0(5)	6 1 5 1	1.596	1.605	51(10)3	0(10) 4 0(10,1)1	1	1.602	0(10) 6 0(10) 5	1	1.607	1.608
516 534	0( 5) 7	1	1.613	0( 5)	8 1	1.601	1.607	STEINIE	0(10) 7	i	1.613	0(10) 8	ĩ	1.605	1.607
510 574	0(5)9	i	1.610	0(5)	ni i	1.601	1.007	51(10)4	0(10) 9	1	1.608	0(10)11	1	1.606	1.003
51( 5)5	0( 4,5)2 0( 5) 9	1	1.605	0(5) 0(5)	6 1 10 1	1.603 1.599	1.605	\$1(10)5	0(10,1)2 0(10) 9	1	1.609	0(10) 6 0(10)10	1 1	1.606	1.609
51( 5)6	0( 5,6)4	1	1.609	0(5)	7 2	1.608	1.606	51(10)6	0(10,9)4	1	1.598	0(10) 7	2	1.603	1.606
SIC 5)7	0( 4,5)3	i	1.606	0( 5)	1 6	1.613	1.606	\$1(10)7	0(10,1)3	i	1.611	0(10) 1	6	1.604	1.603
51( 5)8	0( 5) 4	1	1.594	0(5) 0(5)	26	1.596	1.603	\$1(10)8	0(10) 4	1	1.601	0(10)11	1 6	1.598	1.606
	0(5)3	6	1.614	0(5)	12 1	1.599		l	0(10) 3	6	1.613	0(10)12	1	1.601	

The average Si–O values for each of the 80 tetrahedra range from 1.602 to 1.611 Å. The O–O distances and O–Si–O angles of the tetrahedra are given in Table 3; the 480 O–O distances range from 2.57 to 2.69 Å with an average value of 2.63 Å. The accuracy of the atomic coordinates was not estimated from the solution to the normal equations, since the inverse-matrix elements

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were not calculated by the version of the CG procedure then being used. The pseudosymmetry present, but not imposed on the structure by the space group or the distance restraints, suggests, along with the low R value and good stereochemistry, that the standard deviations in atomic positions are  $\sim 0.02$  Å. Taking into account the distance constraints applied and the expected and Table 3. O-O distances (Å) and O-Si-O angles (°) within tetrahedra in low tridymite

Conventions are as in Table 2. Estimated standard deviation of an individual distance is approximately 0.02 Å.

ATON 1	ATOR 2	SYM2	ATOM 3	SYM3	DIS	ANG	AVG	ATON 1	ATOM 2	SYN2	ATON 3	SYN3	DIS	ANG	AVG
SIC 1)1	0( 1,2)1	1	0(1)1	1	2.614	108.4	2.628	51( 2)6	0( 1,2)4	1	0(2)7	2	2.620	109.4	2.625
	0( 1,2)1	1	0(1)2	1	2.622	109.0			0( 1,2)4	ĩ	0( 2)10	1	2.624	108.7	
	0( 1,2)1	1	0( 1) 5	1	2.614	109.0			0( 1,2)4	1	0( 2)12	1	2.625	109.8	
		1	0(1) Z	1	2.617	108.5			0(2)7	2	0( 2)10	1	2.592	107.0	
	0(1)2	i	0(1)5	. i	2.644	110.6		1	0(2)7	2	0(2)12	1	2.602	107.8	
SIC 1)2	0(1,2)2	i	0(1)3	i	2.626	110.4	2.616	51( 2)7	0(2,3)3	i	0(2)1	6	2.633	109.6	2.630
	0( 1,2)2	1	0(1)4	1	2.585	107.5			0( 2,3)3	1	0( 2) 4	7	2.624	109.1	
	0( 1,2)2	1	0(1)6	1	2.595	109.0			0( 2,3)3	L	0( 2)11	1	2.629	109.0	
	9(1)3	1			2.661	108.2				6	0(2)4		2.607	108.4	
	0(1)4	i	0(1)6	i	2.619	109.3			0(2)4	ž	0( 2)11	î	2.673	112.2	
SIC 1)3	0(10,1)2	11	0(1)5	1	2.627	109.2	2.628	51( 2)8	0( 2,3)4	1	0( 2) 2	6	2.633	110.2	2.625
	0(10,1)2	11	0(1)7	1	2.644	110.9			0( 2,3)4	1	0(2)3	6	2.628	109.8	
	0(1)5	11		1	2.622	108.4			0(2,3)4	1	0(2)12	1	2.660	107.9	
	0(1)5	i	0(1)8	i	2.582	106.2			0(2)2	6	0( 2)12	ĭ	2.603	108.0	
	0( 1) 7	1	0(1)8	1	2.679	112.8			0(2)3	6	0( 2)12	1	2.621	109.0	
SIC 134	0(1,2)3	1	0(1)8	1	2.613	108.2	2.629	51(3)1	0( 3,4)1	1	0(3)1	1	2.659	112.2	2.622
	0(1,2)3	1	6(1) 9	' <u>1</u>	2.632	111+1			P( 3,4)1	1	0(3)2	1	2.614	109.3	
	0(1)8	ī	0(1) 9	i	2.654	110.8			0(3)1	- 1	0(3)2	i	2.598	107.6	
	0(1)8	1	0( 1)11	1	2.606	108.0		1	0( 3) 1	1	0( 3) 5	1	2.644	110.6	
\$17.136	0(1)9	1	0(1)11	. 1	2.614	108.8			0(_3)_2	1	0(3)5	1	2.618	108.9	
31( 1))	0(10,1)1	9	0(1)9	i	2.644	110.3	2.021		0( 3+4)2	1	0(3)3	1	2.651	109.0	2.622
	0(10,1)1	9	0( 1)10	i	2.616	108.4		1	0( 3,4)2	î	0(3)6	i	2.605	108.5	
	0(1)6	1	0( 1) 9	1	2.624	109.6		]	<b>đ(</b> 3) 3	1	0(3)4	1	2.614	108.9	
	0(1)6	1	0( 1)10	1	2.586	107.1		1	0(3)3	1	0(3)6	1	2.652	111.4	
SIC 1)6	0(1,2)4	i	0(1)7	2	2.624	109.4	2.624	516 333	0(2,3)	1	0(3)6	1	2.594	107+6	2.425
	0( 1,2)4	ī	0( 1)10	i	2.611	108.3			0( 2,3)1	i	0(3) 7	i	2.625	109.4	2.027
	0( 1,2)4	1	0( 1)12	1	2.653	111.8		1	0( 2,3)1	1	0(3)8	1	2.596	107.5	
		2	0(1)10		2.652	110.7			0(3)5	1	0(3)7	1	2.637	110.3	
	0(1)10	1	0(1)12	1	2.599	108.0			0(3)5	1	0(3)8	1	2.640	110.4	
SIC 1)7	0(10,1)4	-1Î	0(1)1	6	2.629	109.6	2.627	51( 3)4	3(3,4)3	1	0(3)8	i	2.625	109.4	2.624
	0(10,1)4	11	0(1)4	7	2.625	109-2			MC 3,4)3	1	0(3)9	1	2.620	108.9	
	0(10,1)4	11		1	2.617	109.3			0( 3,4)3	1	0( 3)11	1	2.616	109.1	
		6	0(1)1	1	2.618	109.2			0(3)8	1	0(3)9	1	2.635	108.6	
	0(1)4	7	0( 1)11	i	2.661	111.7		ł	0(3)9	ī	0(3)11	i	2.634	110.3	
SIC 1)8	0(10,1)3	9	0( 1) 2	6	2.638	110.0	2.622	51( 3)5	0( 2,3)2	ī	0(3)6	1	2.623	109.3	2.625
	0(10,1)3	9	0(1)3	6	2.630	110.0			0( 2,3)2	4	0(3)9	1	2.628	109.4	
	0(1)2	6	0(1)12		2.620	108.7		1	D( 2,3)2	1	0(3)10	1	2.581	107.0	
	0(1)2	ő	0( 1)12	ĩ	2.644	110.8		1	0(3)6	1	0(3)10	i	2.647	111.1	
	0( 1) 3	6	0( 1)12	1	2.606	108.9		1	0( 3) 9	ī	0( 3)10	1	2.638	110.3	
51( 2)1	0( 1,2)1	1	0(2)1	1	2.643	110.9	2.625	SIC 336	0( 3,4)4	1	0( 3) 7	2	2.591	107.2	2.625
	0(1,2)1	1	0(2)2		2.641	110.7		1	0( 3,4)4	1	0(3)10	1	2.627	109.4	
	0(2)1	ī	0(2)2	i	2.615	108.9		1	0(3)7		0( 3)10	÷	2.638	110.6	
	0(2)1	1	0( 2) 5	1	2.600	107.7		{	0( 3) 7	2	0( 3)12	ĩ	2.636	110.3	
	0(2)2	1	0(2) 9	1	2.607	108.0		1	0( 3)10	1	0( 3)12	1	2.641	110.7	
51( 2)2	0(1,2)2	1	6(2)		2.581	107.7	2.621	510 357	TC 2+333	1	0(3)1	6	2.637	110.0	2.623
	0( 1,2)2	1	0(2)6	i	2.647	111.7		ł	0(2,3)3		0(3)1		2.651	109.2	
	0(2)3	1	0(2)4	1	2.639	109.9			0(3)1	6	0(3)4	7	2.610	108.5	
	0(2)3	1	0(2)6	1	2.601	107.5		1	0( 3) 1	6	0( 3)11	1	2.577	106.7	
\$1( 2)3	0(2)4	1	0(2)0		2+632	109.8	2 4 2 0	1 576 330	0(3)4	!	0(3)11	1	2.646	111.2	
511 275	0(2,3)1	i	0(2)1	i	2.621	109.2	2.027	310 370	0( 2,3)4	1	0(3)3	6	2.6093	108.6	2.023
	0( 2,3)1	1	0(2)8	i i	2.623	108.7		1	0( 2,3)4	ī	0( 3)12	ĭ	2.615	109.2	
	0(2)5	1	0(2)7	1	2.647	111.0		ļ	0( 3) 2	6	0(3)3	6	2.619	109.0	
	0(2)5	1	n(2)8		2.673	107.6		1	0(3)2	6	0( 3)12	1	2.655	111.4	
51( 2)4	0(1,2)3	i	0(2)8	i	2.630	109.4	2.628	516 431	0(3,4)1	1	0(4)1	1	2.599	107.8	2-622
	0( 1,2)3	1	0(2)9	) 1	2.590	107.5			0( 3,4)1	ĩ	0( 4) 2	ī	2.617	109.2	
	0( 1,2)3	1	0( 2)11	1	2.637	109.8			0( 3,4)1	1	0( 4) 5	1	2.599	107.8	
	0(2)8	1	0(2)9	1	2.661	111.7		1	0(4)1	1	0(4)2	1	2.665	112.4	
	0(2)9	î	0( 2)11	i	2.633	110.0		1	0(4)2	1	0(4)5	1	2.621	109.4	
\$1( 2)5	0( 2,3)2	1	0(2)6	1	2.628	109.3	2.627	STC 4)2	0( 3,4)2	ī	0( 4) 3	ī	2.605	108.0	2.629
	0( 2,3)2	1	0(2)9	1	2.613	109.0		1	0( 3,4)2	1	0( 4) 4	1	2.636	109.7	
	0(2,3)2	1	0(2)10	1	2.602	111.8		l	0(3+4)2	1	0(4)6	1	2.617	108.6	
	0(2)6	ī	0( 2)10	i i	2.603	107.7		1	0(4)3	i	0(4)6	i	2.681	112.8	
	0(2)9	1	0( 2)10	) 1	2.632	110.0		1	0( 4) 4	ĩ	0( 4) 6	ĩ	2.620	108.9	

experimental distributions of interatomic distances, we estimate that the standard deviations for individual distances are as follows: Si-O 0.005; O-O 0.02; Si-Si 0.03 Å. The isotropic *B* for the Si atoms refined to a value of 0.61 Å<sup>2</sup>. The refined values for the axes describing the anisotropic ellipsoid of the O atoms are 0.60 Å<sup>2</sup> for the axis parallel to the line connecting the bridged Si atoms, 2.20 Å<sup>2</sup> for the axis perpendicular to

the Si–O–Si plane, and  $1.65 \text{ Å}^2$  for the third axis. The Si–Si distances and Si–O–Si angles arc given in Table 4, with the symmetry transformations used to generate Tables 2–4 given in Table 5.

The structure will be discussed in terms of layers of tetrahedra parallel to (001). Since the structure is a framework, such a description, of course, is somewhat arbitrary. There are, however, several reasons for

ATCH 1	ATON 2	SYM2	ATOM 3	SY#3	DIS	ANG	AVG	ATOR 1	ATON 2	SY#2	ATOM 3	SY#3	DIS	ANG	AVG
SIC 4)3	0( 4,5)1	1	0( 4) 5	1	2.602	108.1	2.625	\$1( 5)8	0( 4,5)4	1	0( 5) 2	6	2.628	110.4	2.617
	0( 4,5)1	1	0(4)7	1	2.629	109.4		1	0( 4,5)4	1	0(5)3	6	2.559	105.8	
	C 4,5)1	1	0(4)8		2.695	109.0			a( 5) 2	4	0( 5)12	4	2.622	108.0	
	0 4) 5	i	0(4)8	i	2.693	114.0			0(5)2	6	0( 5)12	ĭ	2.677	113.3	
	0( 4) 7	ĩ	0(4)8	ī	2.621	109.0			0(5)3	6	0( 5)12	1	2.624	109.6	
51( 4)4	0( 3,4)3	ī	0(4)8	ī	2.617	109.0	2.628	SI( 6)1	0( 5,6)1	1	0( 6) 1	1	2.579	107.0	2.619
	0( 3,4)3	1	0( 4) 9	1	2.619	108.7		{	0( 5,6)1	1	0( 6) Z	1	2.627	109.7	
	0(3,4)3	1	0( 4)11	1	2.666	112.2			0( 5,6)1	1	0(6)5	1	2+610	109+1	
	0(4)8	-	0(4)9	1	2.631	109.9			0(6)1	t i	0( 6) 2	- i	2.628	110.3	
	0(4)9	î	0( 4)11	ī	2.603	107.7		1	0(6)2	ī	0(6)5	i	2.635	110.5	
51( 4)5	0( 4,5)2	1	0( 4) 6	1	2.620	108.8	2.630	51( 6)2	8( 5,6)2	1	0(6)3	1	2.604	108.2	2.623
	0( 4,5)2	1	0(4)9	1	2.612	108.3			0( 5,6)2	1	0( 6) 4	1	2.604	108.0	
	0(4,5)2	1	<b>F( 4)10</b>	1	2.614	108.8			0( 5,6)2	1	0(6)6	1	2.625	109.9	
	0(4)6	1	0(4)10	i	2.644	110.5			0(6)3	1	0(6)6	i	2.632	110.3	
	7(4) 9	ī	0( 4)10	ĩ	2.672	112.1			0( 6) 4	ī	0( 6) 6	ī	2.635	110.3	
51( 4)6	0( 3,4)4	1	- <b>T</b> ( 4) T	Z	2.668	111.9	2.629	51( 6)3	5( 6,7)1	۱	0(6)5	۱	2.643	111.9	2.619
	5( 3,4)4	1	nc 4)10	1	2.601	107.7			0( 6,7)1	1	0( 6) 7	1	2.605	108.8	
	0(3,4)4	1	0( 4)12	1	2.643	110.6			0( 6,7)1	4	0(6)8	1	2.594	107.7	
	0(4)7	2	0(4)12	i	2.604	108.0			0(6)5	1	0(6)8	1	2.610	108.5	
	0( 4)10	ī	0( 4)12	ī	2.627	109.4			0( 6) 7	ī	0( 6) 8	ī	2.615	108.2	
51( 4)7	0( 4,5)3	1	0(4)1	6	2.618	109.5	2.623	SI( 6)4	8( 5,6)3	1	0(6)8	1	2.621	109.0	2.630
	0( 4,5)3	1	0( 4) 4		2.647	111.1			0( 5,6)3	1	0( 6) 9	1	2.654	110.9	
		1	0(4)11	1,	2.607	109.2			0( 5,673	1	0( 6)11	+	2.609	109.1	
	0 4 1	6	0( 4)11	i	2.608	108.5			0(6)8	i	0 6511	i	2.635	109.8	
	0( 4) 4	7	0( 4)11	1	2.612	108.4			0( 6) 9	Ā	0( 6)11	ī	2.639	110.0	
51( 4)8	0( 4,5)4	1	0(4)2	6	2.588	107.7	2.621	51( 6)5	8( 6,7)2	1	0(6)6	1	2.648	112.2	2.619
	0( 4,5)4	1	0(4)3	6	2.584	107.6			7( 6,7)2	1	0(6)9	1	2.587	107.4	
	0(4,5)4	4	0(4)12 0(4)3	1	2.645	112.7			0( 6,1)2	1	0(6)10	-	2.590	107.0	
	0(4)2	6	0 4512	ĭ	2.606	108.1			0(6)6	ī	0 6510	i	2.632	110.3	
	0(4)3	6	0( 4)12	1	2.643	110.6			đ( 6) 9	1	0( 6)10	1	2.602	107.4	
SIC 571	0( 5,6)1	1	e( 5) 1	1	2.641	110.4	2.625	51( 6)6	0( 5,6)4	1	0( 6) 7	2	2.663	111.7	2.626
	0( 5,6)1	1	0(5)2	!	2.590	107.4		1	T( 5,6)4	1	0( 6)10	1	2.602	107.8	
	0( 5,0)(	1	a(5) 2	- 1	2.606	107.9			M( 6) 7	2	0( 6)12		2.615	109.1	
	0( 5) 1	î	0(5)5	î	2.667	112.4			0( 6) 7	Ż	0( 6)12	ī	2.643	110.7	
	0(5)2	1	0(5)5	1	2.606	108.7			0( 6)10	1	0( 6)12	1	2.607	108.3	
SIC 5)2	0( 5+6)2	1	0(5)3	1	2.622	109.3	2.621	51( 6)7	0( 6,7)3	Å.	0( 6) 1	6	2.621	109.5	2.625
	0( 5,6)2	1	0(5)4	1	2.598	107.0			0( 6,7)3	1	0( 6) 4		2.627	109.2	
	0(5)3	ì	0(5)4	i	2.607	108.2			0(6)1	6	0(6)4	÷	2.643	111.0	
	0(5)3	ī	0(5)6	1	2.670	112.9			0( 6) 1	6	0( 6)11	1	2.625	109.8	
	0( 5) 4	1	0(5)6	1	2.619	109.5			0(6)4	1	0( 6)11	1	2.616	108.6	
\$1( 5)3	0(4,5)1	1	0(5)5	1	2.634	110.5	2.620	516 678	0( 6,7)4	1	0(6)2	6	2.624	109.2	2.624
	0( 4,5)1	1	0(5)8	i	2.644	111.2			0( 6.7)4	î	0( 6)12	ĭ	2.596	107.4	
	0( 5) 5	ĩ	0( 5) 7	ī	2.582	106.8			0( 6) 2	6	0( 6) 3	6	2.656	111.8	
	0( 5) 5	1	0( 5) 8	1	2.624	110.0			0( 6) 2	6	0( 6)12	1	2.622	199.3	
	0(5)7	1	0(5)8	1	2.619	109.2			0(6)3	6	0( 6)12	1	2.626	109.7	
31( 3)4	0( 5.6)3	1	n(5) 9	1	2.613	109.5	2.024	310 171	0(7,8)1	1	0(7)2	1	2.635	110.3	2.022
	0( 5+6)3	ī	0( 5)11	ī	2.652	111.4			0( 7,8)1	ĩ	0(7)5	ī	2.596	107.9	
	C(5)8	1	0(5)9	1	2.631	109.8			0(7)1	1	0(7)2	1	2.672	112.5	
	0(5)8	1	0( 5)11	1	2.588	107.6			0(7)1	1	0(7)5	1	2.608	108.4	
CTC 515	0(5)9	1	0(5)11	1	2.635	110.3	2.420	516 732	0(7)2	1		1	2.600	108.2	2.628
31( )))	0( 4,5)2	i	0(5)9	ì	2.628	109.6	21020	310 172	0( 7.8)2	- î	0(7)4	i	2.652	110.7	
	0( 4,5)2	1	0( 5)10	1	2.642	111.1			0( 7,8)2	1	0(7)6	1	2.629	109.6	
	0(5)6	1	7(5)9	1	2.583	107.0			0( 7) 3	1	0( 7) 4	1	2.610	108.2	
	0(5)6	1	0( 5)10	1	2.626	110.2			0(7)3	1	0(7)6	1	2.662	111.8	
\$16 536	0( 5) 9	1	0(5)10	2	2+607	108.6	2.623	516 733	0(7) 4	1	0(7)6	1	2.604	108.6	2.620
510 770	0( 5+6)4	i	0( 5)10	ĩ	2.636	110.3			0( 6,7)1	ī	0(7) 1	î	2.629	109.9	
	0( 5,6)4	i	0( 5)12	ī	2.633	110.1		l	0( 6,7)1	1	0(7)8	1	2.601	108.7	
	0( 5) 7	2	0( 5)10	1	2.648	111.0			0(7)5	1	0( 7) 7	1	2.587	107.1	
	0(5)7	2	0( 5)12	1	2.626	109.7			0(7)5	1	0(7)8	1	2.680	113.4	
\$1( 5)7	0( 4.5)3	1	0(5)12	6	2.515	108.1	2.627	516 714	0(7,8)3	1	0(7)8 0(7)8	1	2.604	109.0	2.624
	0( 4,5)3	i	0( 5) 4	ĭ	2.614	108.9			0( 7,8)3	î	0( 7) 9	i	2.607	108.0	
	0( 4,5)3	1	0( 5)11	1	2.610	109.2			0( 7,8)3	<u>ک</u>	0( 7)11	1	2.672	112.7	
	0(5)1	6	0(5)4	7	2.606	108.0		l	0(7)8	1	0(7)9	1	2.614	108.6	
	0(5)4	7	0 5)11	1	2.674	113.7		I	n(7)9	1	0( 7)11	1	2.595	107.4	
		•		•				1		-		-			

Table 3 (cont.)

choosing this description. The most important among these are (1) the internal symmetry of the layers and how this symmetry is related to possible ambiguities in the structure, (2) the layer-like morphology and cleavage and (3) streaking parallel to  $c^*$  perhaps related to layer-stacking faults. The ten layers in the asymmetric unit are illustrated in Fig. 1.

To an average accuracy of about 0.1 Å, the structure may be thought of as composed of four different types of layers in the sequence illustrated in Fig. 2. Approximate transformations relating the layers are listed in Table 6.

It may be noted that the projections of the basal (inlayer as opposed to those shared by adjacent layers) O atoms on the xy plane are nearly identical for all layers. In order to bring the projections of what will be described as the *B* layers into near-coincidence with the others, it is necessary to rotate about **c** by  $60^{\circ}$  in addition to the necessary reflections and translations concerning x and y.

A most remarkable pseudosymmetry possessed by the A, C and D-type layers is their near-periodicity of a/2, (b/2) (Table 6). As a result, the 32 Si atoms and the 48 in-layer O atoms of the B-type layers are the

Table	3	(cont.)
rabic	2	(00000)

ATON 1	ATON 2 SY	12 AT 0	3 9	5YN 3	DIS	ANG	AVG	ATOM 1	ATOR 2	SYN2	ATON 3	SYM3	DIS	ANG	AVG
SIC 735	0( 6.7)2		1) 6	1	2.620	109.3	2.621	51( 9)3	0( 8,9)1	1	<b>đ(</b> 9) 5	1	2.665	112.4	2.624
51( 1))	0( 6,7)2		1) 9	ĩ	2.626	110.2			0( 8,9)1	1	0( 9) 7	1	2.629	110.1	
	0( 6,7)2	1 0( 1	7)10	1 .	2.587	108.1			0( 8,9)1	1	0(9)8	1	2.695	107.8	
	0(7)6		7310	1	2.588	109.3			0(9)5	i	0(9)8	i	2.593	107.2	
	0(7)9	i oci	7)10	i	2.678	113.2	1		0( 9) 7	1	0( 9) 8	1	2.600	107.6	
SI( 7)6	0( 7,8)4	L 0C	7) 7	2	2.673	112.5	2.625	SI( 9)4	0(10,9)3	1	0(9)8	1	2.661	112.7	2.616
	0(7,8)4		7)10	1	2.628	108+8			0(10,9)3	1	0(9)11	i	2.568	106.2	
	0(7)7	2 80	7)10	i	2.616	109.1	ļ		0(9)8	- I	0(9)9	1	2.690	114+3	
	0( 7) 7	2 0(	7)12	1	2.595	107.1			0( 9) 8	1	0( 9)11	1	2.596	107.2	
	0( 7)10		7)12	1	2.629	109.8	2.624	\$1( 9)5	m( 8.9)2	1	0(9)11	1	2.604	107.7	2.623
51( 7)7	0( 6,7)3	י פר	754	7	2.661	111.8		510	0( 8,9)2	ī	0(9)9	1	2.631	110.5	
	0( 6,7)3	1 0(	7)11	1	2.614	108.8			0( 8,9)2	1	0( 9)10	1	2.702	114.3	
	0(7)1	6 00	7) 4	?	2.628	109.7			0(9)6	1	0(9)9	1	2.515	108.3	
	0(7)1	7 5(	7211	1	2.605	108.1			0(9)9	î	0( 9)10	i	2.610	109.1	
51( 7)8	0( 6,7)4	1 00	7) 2	6	2.626	109.4	2.628	51( 9)6	0(10,9)4	1	0( 9) 7	2	2.607	108.7	2.624
	0( 6,7)4	1 00	7) 3	6	2.620	109.2			0(10,9)4	1	0(9)10	1	2.650	10.6	
	0(6,7)	1 OC	7) 3	6	2.615	108.3			0(9)7	Ż	0( 9)10	i	2.606	108.0	
	0(7)2	6 0(	7)12	ĩ	2.602	107.7			0( 9) 7	2	0( 9)12	1	2.678	113.4	
	9(7)3	6 0(	7)12	1	2.638	110.1	2 4 2 2	577 937	0(9)10	1	0(9)12	1	2.638	107.7	2.627
21( 8)1	0(7,8)1	1 00	8) 2	i	2.589	107.6	2+023	310 771	0( 8,9)3	i	0(9)4	ĩ	2.609	108.5	
	0( 7,8)1	1 00	8) 5	ī	2.594	107.8			0( 8,9)3	1	0( 9)11	1	2.629	110.1	
	0(8)1	1 00	8) 2	1	2.604	108.2			0(9)1	6	0(9)4		2.587	106.6	
	0(8)1	1 00	8) 5	i	2.636	109.9			0(9)4	7	0( 9)11	ī	2.679	113.2	
51( 8)2	0( 7,8)2	i oc	8) 3	i	2.656	111.6	2.624	51( 9)8	0( 8,9)4	1	0( 9) 2	6	2.617	109.6	2.623
	0( 7,8)2	1 00	8) 4	1	2.606	108.1			0( 8,9)4	1	0(9)3	6	2.618	109.5	
	0(7,8)2	1 80	8)6	1	2.607	108.4			n(9)2	6	0(9)	6	2.679	112.8	
	0(8)3	i õč	8) 6	i	2.636	110.4			0( 9) 2	6	0( 9)12	1	2.623	109.2	
	0( 8) 4	1 0(	8) 6	1	2.602	108.0			0( 9) 3	0	0( 9)12	2 1	2.617	108.5	2 4 20
SI( 8)3	0( 8,9)1	1 00	8) 5	1	2.599	107.3	2.629	\$1(10)1	0(10,9)1	1	0(10)	1	2.598	108.9	2.020
	0( 8,9)1	1 00	8) 8	i	2.623	109.1			0(10,9)1	i	0(10) 5	i	2.631	110.0	
	0(8)5	1 90	8) 7	1	2.637	109.9			5(10) 1	1	0(10)	2 1	2.620	109.5	
	0(8)5	1 00	8)8	1	2.648	110.7			0(10) 1	1	0(10) 9	1	2.680	113.5	
516 914	0(8)7	1 00	8) 8	1	2.604	107.7	2.625	51(10)2	0(10,9)2	1	0(10)	3 1	2.619	109.5	2.618
31( 8)4	0( 7,8)3	i oc	8) 9	i	2.609	108.5			0(10,9)2	ī	0(10)	1	2.564	106.2	
	0( 7,8)3	1 00	8)11	1	2.605	108.2			0(10,9)2	1	0(10)	5 1	2.604	108.4	
	0(8)8	1 00	8) 9	1	2.628	109.6		l	0(10) 3		0(10)	• •	2.672	112.8	
	0(8)9	1 00	8)11	i	2.666	112.4			0(10) 4	ī	0(10)	5 1	2.607	108.6	
51( 8)5	0( 8,9)2	1 0(	8) 6	1	2.612	108.4	2.626	\$1(10)3	0(10,1)1	1	0(10)	5 1	2.635	110.1	2.625
	0( 8,9)2	1 00	8) 9	1	2.642	110.0			0(10,1)	1	0(10)	1 1	2.660	108.2	
	0(8)6	1 00	8) 9	i	2.623	109.4			0(10) 5	i	0(10)	<i>i</i> 1	2.587	106.9	
	0(8)6	1 00	8)10	1	2.653	111.6		1	0(10) 5	1	0(10)	3 1	2.656	111.5	
	0(8)9	1 00	8)10	1	2.626	109.6	2 ( 2 2	51(10)4	6(10)		0(10)	5 1 B 1	2.583	108.1	2.617
21( 8)6	0( 7,8)4	1 90	8) 7	1	2.579	108.8	2.023	31(10)4	7(10,9)	, 1 3 1	0(10)	9 1	2.671	113.6	
	0( 7,8)4	1 00	8)12	i	2.623	109.6			0(10,9)	3 1	0(10)1	1 1	2.627	110.9	
	0(8)7	2 0(	8)10	1	2.629	109.7		1	0(10) 8	3 1	0(10)	9 1	2.584	107.9	
	0(8)7	1 00	8)12	1	2.628	110.1			0(10)	i	0(10)1	i i	2.630	109.9	
SIC 8)7	0( 8,9)3	1 00	8) 1	6	2.633	110.1	2.624	\$1(10)5	0(10,1)	2 1	0(10)	61	2.619	109-1	2.627
	0( 8,9)3	1 00	8) 4	?	2.669	112.7			0(10,1)	2 1	0(10)	91	2.595	107+3	
	0(8,9)3 0(8)1	1 D(	8) 4	7	2.599	107.8			0(10)	5 1	0(10)	9 i	2.626	109.4	
	0 ( 8) I	6 0(	8)11	i	2.608	108.2		1	0(10) (	5 I	0(10)1	0 1	2.659	111.0	
	0(8)4	7 00	8)11	1	2.630	109.8	2 4 1 9	5771034	0(10)	<b>7</b>	<b>C(10)1</b>	01	2.602	107.8	2.621
31( 0)0	0( 8,9)4	1 00	8) 3	6	2.606	108.8	2.017	1	0(10,9)	i	0(10)1	0 Ī	2.605	108.2	
	0( 8,9)4	1 0(	8)12	1	2.615	109.6			0(10,9)	4 1	0(10)1	2 1	2.624	109.9	
	0(8)2	6 0(	8) 3	6	2.607	108.2		1	0(10)	12	0(10)1	U 1 2 1	2+598	107+6	
	0(8)2	6 6(	8)12	i	2.584	106.6		1	0(10)1	. i	0(10)1	2 1	2.581	106.4	
SI( 9)1	0(10,9)1	1 00	9) 1	1	2.674	113.0	2.623	\$1(10)7	0(10,1)	3 1	0(10)	1 6	2.603	108.1	2.617
	0(10,9)1	1 0(	9) 2	1	2.613	109.1		1	0(10,1)	3 1	0(10)	• 7	2.562	105+8	
	0(10,9)1	1 00	9) 2	1	2.620	109.3		1	0(10)	16	0(10)	4 7	2.628	110.1	
	ð( 9) i	i õč	9) 5	i	2.617	108.8		1	0(10)	1 6	0(10)1	1 i	2.619	109.8	
	0( 9) 2	1 00	9) 5	1	2.587	106.9		1	0(10)	47	0(10)1	1 1	2.667	113.0	2. 621
51( 9)2	0(10,9)2	1 80	9) 4	1	2.5/1	108-8	2.022	1 31(10)8	0(10,1)	4 1	0(10)	36	2.611	108.3	
	0(10,9)2	i õč	9) 6	i	2.682	114.1		1	0(10,1)	4 ī	0(10)1	2 1	2.618	109.3	
	0(9)3	1 00	9) 4	1	2.596	106.8		1	0(10)	26	0(10)	3 6	2.621	109.4	
	0(9)3	1 00	9) 6	1	2.677	112.9		1	0(10)	36	0(10)1	2 1	2.606	108.4	
									-						

main contributors to the weak uuu reflections. In fact, the *R* value is 11% for the 234 7uu reflections when the model is composed of only the 80 atoms of the *B*-type layers. The *R* for this same subset of uuu data is 9% when all 240 atoms are included in the calculation. in-plane at y = 0.25. These are particularly close for layers 3 and 8. The *B* layers have very approximate centers of symmetry located at the ring centers. The *D* type of layer differs, after appropriate transformations, by only 0.16-0.20 Å from the *C* type, so that at some level they may be considered the same.

There are three other types of pseudosymmetry. Layers 2 and 9 are nearly mirror images of one another across z = 0.25, as are layers 3 and 8, and layers 4 and 7. The A, C and D layers have approximate screw axes

As shown in Fig. 2, the *B* layers are sandwiched between layers with approximate  $\frac{1}{2}$  cell periodicity. If these layers possessed exact  $\frac{1}{2}$  cell periodicity, then

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Conventions are as in Table 2. Estimated standard deviation of an individual distance is approximately 0.03 Å.

ATON 2	ATOM 1	SYM1	ATOM 3	SYM3	DIS	ANG	ATON 2	AT 0H 1	SYM1	ATON 3	5¥#3	0 I S	ANG
0( 1,2)1	51( 1)1	1	SI( 2)1	1	3.156	158.6	<b>đ( 4) 5</b>	SI( 4)1	1	SIC 4)3	1	3.065	144.3
0( 1,2)2	SIC 1)2	1	SI( 2)2	1	3.173	173.2	8( 4) 6	SIC 4)2	1	SIC 4)5	1	3.046	141.9
0( 1,2)3	SI( 1)4	1	SIC 2)4	1	3.106	149.6	0(4)7	SIC 4)3	1	SIC 436	3	3-059	142+7
0( 2.3)1	SI( 2)3	1	SI( 3)3	1	3.075	145.4	0(4)9	SI( 4)4	í	51( 4)5	ī	3.050	141.4
0( 2,3)2	SIC 2)5	ī	SI( 3)5	ī	3.105	149.6	OC 4)10	SIC 4)5	1	SIC 4)6	1	3.118	151.3
0( 2,3)3	SI( 2)7	1	SIC 3)7	1	3.105	148.8	0( 4)11	SIC 4)4	1	SIC 4)7	1	3.080	146.7
0(2,3)4 0(3,4)1	SIC 208	1	SI( 3)8	1	3.131	154.7	0(-4)12 0(-5)1	SI( 476	1	SIC 470	5	3.074	144.8
0( 3,4)2	SI( 3)2	i	SIC 4)2	i	3.082	146.4	0( 5) 2	51( 5)1	i	SIC 538	5	3.056	143.5
0( 3,4)3	SIC 3)4	1	SIC 4)4	1	3.081	146.8	0(5)3	SIC 5)2	1	SI( 5)8	5	3.074	145.3
0( 3,4)4	SIC 3)6	1	SIC 4)6	1	3.091	147.1	0(5)4	51( 5)2	1	SIC 537	1	3.146	192.7
0( 4,5)2	SI( 4)5	1	SI( 5)5	i	3.094	148.6	0(5)6	51( 5)2	i	51( 5)5	ī	3.140	158.1
0( 4,5)3	SI( 4)7	1	SI( 5)7	1	3.144	157.6	0( 5) 7	SIC 5)3	1	SIC 536	3	3.055	143.1
0( 4,5)4	SI( 4)8	1	SI( 5)8	1	3.148	162.6	0(5)8	SIC 533	1	51( 5)4	1	3.108	151.5
3( 5,6)2	SIC 5)2	i	SIC 6)2	i	3.080	146.8	0( 5)10	SIC 535	1	51( 5)6	i	3.104	151.4
0( 5,6)3	SI( 5)4	1	SI( 6)4	1	3.089	147.2	0( 5)11	SIC 5)4	1	SIC 5)7	1	3.142	158.8
0( 5,6)4	SI( 5)6	1	SI( 6)6	1	3.082	146.5	0( 5)12	SIC 5)6	1	SIC 5)8	1	3.147	158.6
0( 6,7)1	51( 6)3	1	51(7)3	1	3.160	166.3	0(6)1	51( 6)1	1	51( 6)/	5	3.067	145.1
0( 6.7)3	SIC 6)7	1	SIC 7)7	i	3.103	149.5	0(6)3	51( 6)2	i	51( 6)8	ś	3.083	147.7
0( 6,7)4	SI( 6)8	1	SIC 7)8	1	3.081	147.0	0(6)4	SI( 6)2	1	SI( 6)7	4	3.057	143.1
0( 7,8)1	SIC 7)1	1	SIC 8)1	1	3.108	152.6	0( 6) 5	SIC 6)1	1	51( 6)3	1	3.125	155.3
0( 7,8)2	51(7)2	1	51( 8)2	1	3.052	147.4	0(6)0	51( 6)2	1	51( 6)5	3	3.051	142.5
0( 7,8)4	SIC 7)6	i	51( 8)6	i	3.073	145.8	0(6)8	51( 6)3	i	51( 6)4	i	3.051	142.3
0( 8,9)1	SI( 8)3	1	SI( 9)3	ī	3.078	145.9	0(6)9	SI( 6)4	1	SIC 6)5	1	3.065	142.8
0( 8,9)2	SI( 8)5	1	SIC 9)5	1	3.108	149.1	0( 6)10	SIC 6)5	1	51( 6)6	1	3.064	144.2
0( 8,9)3	SIC 8)7	1	51( 9)7	1	3.085	148.0	0(6)11	51( 6)4	1	51( 6)7	1	3.096	151+1
0(10.9)1	SIC 9)1	ì	51(10)1	1	3.101	150.0	0(7)1	SI( 7)1	i	\$1( 7)7	5	3.068	146.0
0(10,9)2	SI( 9)2	ī	\$1(10)2	ī	3.085	149.3	0(7)2	\$1( 7)1	1	51( 7)8	5	3.066	143.8
0(10,9)3	SI( 9)4	1	SI(10)4	1	3.157	167.6	0(7)3	SIC 7)2	1	SIC 7)8	5	3.086	147.1
0(10,9)4	51( 9)6	10	51(10)6	1	3.126	154.7		51(7)2	1	51( 7)3	1	3.086	147.2
0(10,1)2	SIC 133		SI(10)5	1	3.082	146.7	0(7)6	51( 7)2	i	51( 7)5	i	3.054	142.5
7(10,1)3	SIC 1)8	10	SI(10)7	1	3.090	148.0	0(7)7	SIC 7)3	1	51( 7)6	3	3.059	143.3
0(10,1)4	SI( 1)7	8	\$1(10)8	1	3.091	147.9	0(7)8	SIC 733	1	SIC 7)4	1	3.089	149.0
0(1) 1		1	SIC 1)8	5	3.051	141.8	0(7)10	51( 7)5	1	51( 7)6	i	3.103	151.6
0(1)3	SIC 1)2	ì	51( 1)8	5	3.041	142.2	0( 7)11	SI( 7)4	ī	51( 7)7	1	3.095	148.8
0(1)4	SIC 1)2	1	SIC 1)7	4	3.057	142.4	0(7)12	SI( 7)6	1	510 738	1	3.067	144.2
0(1)5		1	51(1)3	1	3.167	159.6	0(8)1		1	51( 8)7	5	3.070	144.8
0(1)7	SIC 1)2	1	SIC 1)6	3	3.094	148.9	0(8)3	51( 8)2	î	51( 8)8	ś	3.064	144.4
0(1)8	SI( 1)3	1	SIC 1)4	1	3.056	142.2	0(8)4	SIC 8)2	1	51( 8)7	4	3.065	144.7
0(1)9	SI( 1)4	1	SIC 1)5	1	3.089	147.3	0(8)5	SIC 8)1	1	SIC 833	1	3.104	148.5
0( 1)10	51(1)5	1	51(1)6	1	3.144	157.1	0(8)0	51( 8)3	1	51( 8)5	3	3.079	192.1
0(1)12	SI( 1)6	i	SIC 1)8	i	3 • 136	158.5	0(8)8	51( 8)3	i	SI( 8)4	í	3.089	147.3
0(2)1	SI( 2)1	1	51( 2)7	5	3.055	143.8	0(8)9	SI( 9)4	1	SIC 8)5	1	3.089	148.1
0(2)2	SIC 2)1	1	SI( 2)8	5	3.092	148.5		SI( 8)5	1	51(8)6	1	3.098	149.7
0(2)3	SIC 2)2	1	SIC 2)7	4	3.074	146.1	0( 8)12	51( 8)6	i	SIC 838	î	3.127	154.0
0(2)5	SI( 2)1	1	SI( 2)3	1	3.081	146.4	0(9)1	SI( 9)1	1	SIC 9)7	5	3.063	143.5
0(2)6	SIC 2)2	1	SI( 2)5	1	3.056	142.7	0(9)2	SI( 9)1	1	51( 9)8	5	3.087	147.7
0(2)7	51(2)3		51(2)6	1	3-061	149.0	0(9) 5	51( 9)2	1	SIC 9)7	4	3.099	148.7
0(2)9	51(2)4	i	51( 2)5	i	3.142	156.8	0( 9) 5	\$1( 9)1	i	51( 9)3	i	3.093	148.4
0( 2)10	SI( 2)5	1	51( 2)6	1	3.053	141.9	0(9)6	SIC 9)2	1	51( 9)5	1	3.061	144.1
0( 2)11	SIC 2)4	1	SIC 2)7	1	3.053	142.3	0(9)7	51( 9)3	1	51( 9)6	3	3.067	146.5
d( 3) 1	SI( 3)1	1	SIC 278	5	3.051	142.8	0 9 9	SIC 9)4	i	51( 9)5	î	3.087	150.4
0(3)2	SIC 3)1	i	51( 3)8	5	3.060	143.8	0( 9)10	SIC 9)5	ī	\$1( 9)6	1	3.053	142.1
0(3)3	SIC 3)2	1	SIC 338	5	3.065	145.0	0( 9)11	SIC 9)4	1	SIC 9)7	1	3.049	142.3
0(3)4		1		4	3.066	145.1		51( 9)6	1	51( 9)8	1	3.064	144.7
0(3)6	SIC 3)2	i	\$1( 3)5	i 1	3.119	152.7	0(10) 2	51(10)1	i	51(10)8	5	3.106	151.4
0( 3) 7	SIC 3)3	1	SIC 3)6	3	3.092	149.1	0(10) 3	SI(10)2	1	\$1(10)8	5	3.061	144.5
0(3)8	SIC 333	1	SIC 334	1	3.094	148.9	0(10) 4	SI(10)2	1	SI(10)7	4	3.094	149.9
0(3)10	SI( 3)5	1	51(3)5	1	3.091	148.7	0(10) 6	SI(10)1	1	SI(10)5	1	3.111	151.2
0( 3)11	51( 3)4	i	SI( 3)7	i	3.130	155.6	0(10) 7	SI(10)3	i	\$1(10)6	3	3.074	145.9
0( 3)12	SIC 336	1	SIC 338	1	3.106	150.3	0(10) 8	SI(10)3	1	SI(10)4	1	3.061	143.9
O( 4) 1	51( 4)1	1	51(4)7	5	3.041	149.3	0(10) 9	51(10)4	1	51(10)5	1	3.044	144.8
0(4)3	SI( 4)2	i	SI( 4)8	5	3.083	146.7	0(10)11	SI(10)4	i	\$1(10)7	i	3.115	153.1
0( 4) 4	SIC 4)2	1	SIC 4)7	4	3.058	143.4	0(10)12	SI(10)6	1	SI(10)8	1	3 • 125	153.9

shifting the *B* layers by  $\frac{1}{2}$  in *x* or *y* (in reality, of course, it would be a series of rotations retaining bonding topology that would have the same result as a shift) would result in exactly the same stereochemistry between the adjacent layers. Evidently, it is because the  $\frac{1}{2}$  cell periodicity is not exact that the *B* layers order.

It may be helpful to consider the structure as composed of ABCBA layer sequences alternating with DBCBD layer sequences. As indicated in Table 6, the AD,DA junctions serve approximately as alternating mirror planes and glide planes with respect to the BCB sequences. It is worth emphasizing that the two BCB



Fig. 1. The structure of low tridymite, shown in layers perpendicular to **c**; **a** across, **b**/2 down. Large numbers (1-8) are Si atoms (dark circles) numbered identically in each layer, corresponding to Tables 1-3. Small unprimed numbers refer to O atoms within the rings which define the layers; small primed numbers refer to O atoms shared with adjacent layers. These also are numbered identically within each layer corresponding to Tables 1-3. The authors have found transparent overlays of the layers useful in viewing the structure.

sequences in the asymmetric unit are only 0.08 Å, on the average, from superimposable; whereas the two A layers and the two D layers are superimposable to about 0.12 Å. Indeed the 5-6 junction and the 10-1 junction are somewhat different, with the former resulting in an approximate mirror relating the BCB sequences and the latter resulting in the near-glide-plane relation.

Models composed of various  $\frac{1}{2}$  cell shifts of the *B* layers were used to calculate *R* values for the previously mentioned 7*uu* data subset. The results are summarized in Table 7. Shifting layers 4 and 9 by  $\frac{1}{2}a$  resulted in an *R* of 14% vs 11% for the refined model.



Fig. 1 (cont.)



Fig. 2. Sequence of layers in the structure of low tridymite. The layers are numbered as in Fig. 1 and Tables 1-3; letters refer to the layer type as described in the text.

Table	5.	Symmetry	transformations	used	to	generate
			Tables 2–4			

Table	6.	Pse	udosymme	etric	transform	nations	relating
layers	in	low	tridymite	and	average	deviatio	ons from
<b>S1</b>	ире	rimp	osability f	or di <u>f</u>	ferent pa	irs of lay	vers

Symmetry		Translations	
number	а	b	С
1	0	0	0
2	1	0	0
3	-1	0	0
4	$\frac{1}{2}$	$-\frac{1}{2}$	0
5	$-\frac{1}{2}$	$-\frac{1}{2}$	0
6	$\frac{\overline{1}}{2}$	$\frac{\overline{1}}{2}$	0
7		$\frac{1}{2}$	0
8	12	Ō	1
9	1/2	0	$-\frac{1}{2}$
10	$-\frac{1}{2}$	0	1/2
11	$-\frac{1}{2}$	0	- <del>1</del>

These two models represent different levels of refinement of the same structure, and differ only by interchanging the positions of the pseudo-mirrors and pseudo-glides relating the BCB sequences. Shifting layers 4 and 7 is similarly the same as shifting 8 and 9. This distinctly different structure refines to an R of 8.4%. As summarized in Table 7, various combinations of different structure give remarkably low Rvalues. Of all the possibilities, only the 4,9 shift and the 4,7 shift were refined; the former before its relation to the first structure was realized (it refined to the same Rof 6.4%) and the latter because it possessed the next lowest R for the 7uu data.

An additional property of the B layers is the ability of a chain of tetrahedra in such a layer to change drastically its conformation without altering its linkage with the remainder of the three-dimensional network. In Fig. 3 is depicted layer 9' at an R of 9%. A difference map indicated the true conformation to be that formed

Layer	Туре	Ar trar	proximate Isformatio	e n	Average a/2 rep	deviation beat within layer	from the
1	4	v	,,	-		0 00 1	
2	R	л,	<i>y</i> ,	2		0.08 A	
2	C	л,	<i>y</i> ,	Z		0.91	
3	D D	х,	У,	Z		0.12	
4	В	$x - \frac{1}{4}$	$-y + \frac{1}{2}$	$-z + \frac{1}{4}$		0.94	
S	A	$x - \frac{1}{4}$ ,	$-y + \frac{1}{2}$	$-z + \frac{1}{4}$		0.08	
6	D	х,	у,	Ζ		0.10	
7	B	$x - \frac{1}{4}$ ,	$-y + \frac{1}{2}$ ,	$z - \frac{1}{4}$		0.88	
8	С	х,	у,	$-z + \frac{1}{2}$		0.06	
9	В	х,	у,	$-z + \frac{1}{2}$		0.90	
10	D	$x - \frac{1}{4}$ ,	$-y + \frac{1}{2}$ ,	$-z + \frac{3}{4}$		0.08	
		А	verage dev	viation			
		Layers	between la	yers	Type(s)		
		1-5	0.12	Å	A		
		6-10	0.11		D		
		3-8	0.07		С		
		2-4	0.05		В		
		2–7	0.06		В		
		2–9	0.08		В		
		4-7	0.08		B		
	~	3-10	0.16		C-D		
	•	1–3	0.52		A-C		

by rotations about the darkened tetrahedra edges. When considering transitions and disorders brought on by heating, it may be worth considering such rotations.

# Comparison with monoclinic low tridymite

The crystal structure of low tridymite, from material typical of terrestrial occurrences, is reported here. It

# Table 7. Conventional unweighted R values for the234 7uu reflections for various combinations of B-typelayer shifts

Layers are numbered as in Fig. 2. Only the 80 atoms in *B*-type layers were included.

Layer 2 shift	Layer 4 shift	Layer 7 shift	Layer 9 shift	R
0	0	0	0	0.110
0	1/2	0	1 <u>7</u>	0.139
0	1 1	17	Ō	0.161
0	Õ	1/2	1/2	0.170
0	0	Ō	<u> </u>	0.202
0	1/2	$\frac{1}{2}$	1 1	0.212
0	1 1	Ō	Ō	0.318
0	Õ	$\frac{1}{2}$	0	0.331



Fig. 3. Layer 9': an incorrect conformation of layer 9 resulting from rotations about the darkened tetrahedral edges. Labeling conventions are the same as in Fig. 1.

differs significantly from the structure of meteoritic and synthetic 'low-tridymites' as described by Dollase & Baur (1976) and Kato & Nukui (1976). Gardner & Appleman (1974) pointed out that even the corresponding orthorhombic subcells of these two types of tridymite show a volume difference of approximately  $4 \cdot 1\%$ . The terrestrial tridymite which we studied has ten crystallographically distinct layers, and the sixmembered rings of tetrahedra which form these layers are all 'oval' (Fig. 1). In contrast, meteoritic tridymite has only two crystallographically distinct layers, and these layers contain  $\frac{1}{4}$  'oval' and  $\frac{2}{4}$  'ditrigonal' rings (Dollase & Baur, 1976). The reasons for the much greater complexity of the terrestrial material remain unclear. Some pertinent comparisons with the structure of Dollase & Baur follow.

The interatomic distances and angles in the two structures are similar. Average Si–O distances for individual tetrahedra range from 1.602 to 1.611 Å (terrestrial) and from 1.587 to 1.616 Å (meteoritic), but the significance of the differences is not certain, in view of the constraints applied in our refinement. Si–Si nearest-neighbor distances range from 3.03 to 3.17 Å (terrestrial, Table 4), and from 3.04 to 3.18 Å (meteoritic). The Si–O–Si angles correspondingly range from 139.7 to  $173.2^{\circ}$  with an average of  $148.3^{\circ}$ (terrestrial) and from 141.7 to  $178.5^{\circ}$  with an average of  $149.8^{\circ}$  (meteoritic). In both structures, there is only one very large Si–O–Si angle. Only five of the Si–O–Si angles are greater than  $160^{\circ}$  in the terrestrial structure. All of these large Si–O–Si angles are between layers rather than within a layer. The average Si–O–Si angles found in these structures are close to those observed in other SiO<sub>2</sub> polymorphs (O'Keeffe & Hyde, 1976).

The average isotropic temperature factors are significantly higher in the meteoritic structure than in the terrestrial tridymite:  $B_{av}(Si) = 0.61$  (terrestrial), 0.93 (meteoritic);  $B_{av}(O) = 1.48$  (terrestrial), 1.93 Å<sup>2</sup> (meteoritic). It seems possible that the meteoritic and synthetic compounds represent a higher-temperature form of tridymite with more positional disorder. It is noteworthy that, as pointed out by Dollase & Baur (1976), individual cristobalite- and tridymite-like structures generally have only one type of ring configuration: either all oval or all ditrigonal. Low cristobalite has all oval rings, high cristobalite may have all ditrigonal rings. By tenuous analogy, meteoritic 'low-tridymite', with  $\frac{1}{3}$  oval and  $\frac{2}{3}$  ditrigonal rings, may represent a higher-temperature modification than terrestrial low tridymite, with all oval rings.

No evidence was found for ordering of aluminum into particular tetrahedral sites. Indeed, the chemical analysis shows slightly less than one of the 80 tetrahedral atoms to be aluminum. At the current level of refinement, we are unable to detect any ordering of atoms within the structural cavities. Our results fail to show any effects of impurities in stabilizing the tridymite structure.

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# Structure Cristalline du Polysulfure de Lanthane LaS,

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Lanthanum disulphide,  $LaS_{22}$  is orthorhombic, space group *Pnma*, with unit-cell constants a = 8.131 (5), b = 16.34 (1), c = 4.142 (2) Å and Z = 8. The crystal structure has been determined from Syntex automatic diffractometer data, using Patterson and Fourier syntheses, and refined by the least-squares method to a final R of 0.045. Sheets of (LaS), parallel to the [100] axis, are formed from La<sub>3</sub>S pyramids; between these layers, planes of S atoms are inserted, formed from S–S pairs in which the interatomic distance (2.104 Å) indicates a covalent bond. The structure is related to a distorted arrangement of anti-Fe<sub>2</sub>As type.

### Introduction

De nombreux travaux ont déjà été réalisés sur les polysulfures des terres rares  $LnS_2$ , mais l'accord ne s'est pas encore fait sur leur maille cristalline.

Flahaut, Guittard & Patrie (1959) ont décrit, il y a longtemps, d'après des diagrammes de Debye et Scherrer, les polysulfures  $LnS_2$  (Ln = La, Ce, Pr, Nd et Sm) comme ayant une maille cristalline de symétrie cubique. Marcon & Pascard (1968) indiquent que le polysulfure de cérium  $CeS_2$  est isotype du polyséléniure  $CeSe_2$ , de symétrie monoclinique; mais leur détermination de la structure du polysélénieure  $CeSe_2$ , effectuée sur monocristal, n'a pu les conduire qu'à une valeur du *R* des cristallographes égale à 0,14.

Yanagisawa & Kume (1973) préparent, sous haute pression, un polysulfure de cérium dont la maille cristalline présente une symétrie orthorhombique, et donnent pour constantes du réseau: a = 8,11; b =16,20; c = 4,09 Å. Mais ils n'ont ni déterminé le groupe spatial, ni résolu la structure. Seule a été déterminée par une étude sur monocristal, jusqu'à une valeur finale de R égale à 0,042, la structure d'un polysulfure d'ytterbium YbS<sub>2</sub> (Teske, 1974); cette phase, différente du composé que nous avons étudié, possède une maille cristalline de symétrie monoclinique, dont les constantes sont:  $a = 9,32_6$ ; b = $4,75_4$ ;  $c = 9,02_5$  Å;  $\beta = 118,3_9^{\circ}$ .

## Préparation

Nous préparons le polysulfure de lanthane  $LaS_2$  par chauffage, à 700 °C en ampoule de silice vide d'air, d'un mélange de sesquisulfure de lanthane  $La_2S_3$  et de soufre en quantité stoechiométrique.

La poudre cristalline ainsi obtenue, mélangée à cinq parties de l'eutectique KI-KCl, est chauffée, toujours en ampoule de silice, à 750 °C pendant trois semaines, puis refroidie lentement. Après lavage à l'eau froide, on recueille de nombreux cristaux transparents, de couleur rouge orangé, ayant la forme de plaquettes. A partir de